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Combustion synthesis of high-entropy alloys and thermoelectric materials

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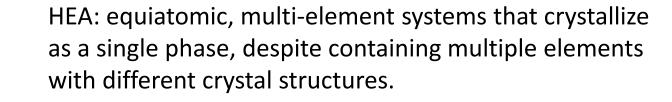
high-entropy alloys

Outline

1. Introduction

- 2. Experimental
- 3. Results and discussion
- 4. Conclusions

High Entroy Alloys (HEA)



$\mathbf{G} = \mathbf{H}\text{-}\mathbf{T}\mathbf{S}, \ \mathbf{S}\uparrow \to \mathbf{G}\downarrow$

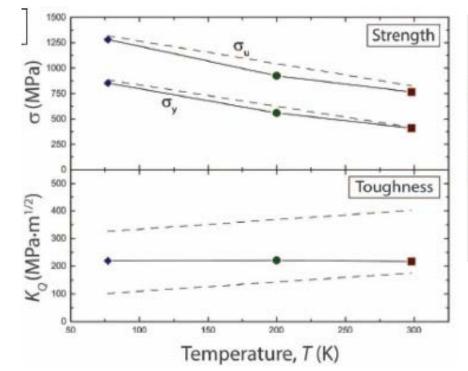
The configurational entropy contribution to the total free energy in alloys with five or more major elements may stabilize the solid-solution state relative to multiphase microstructures.

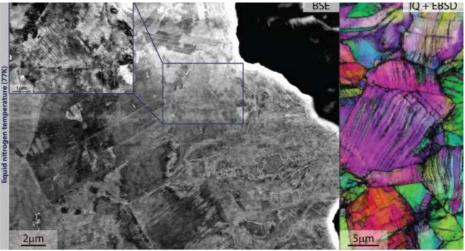
The microstructure of HEA is often characterized with lattice distortion and nano-precipitates, which contributes to interesting mechanical properties.

Conventional Alloys

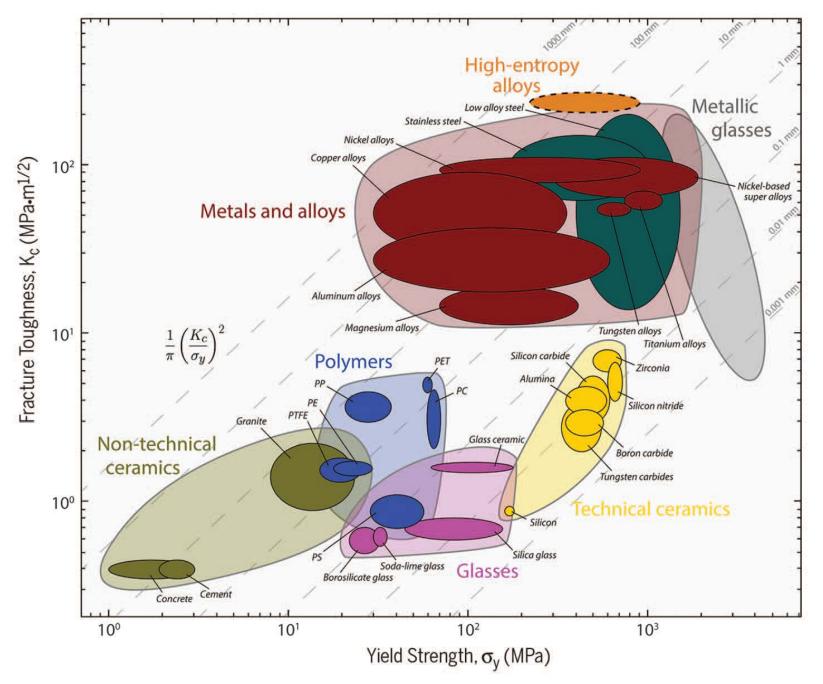
CrMnFeCoNi HEA: Improved properties at Low T

The mechanical properties of CrMnFeCoNi HEA actually improve at cryogenic temperatures. This is attributed to a transition from planar-slip dislocation activity at room temperature to deformation by mechanical nanotwinning with decreasing temperature, which results in continuous steady strain hardening.





Bernd Gludovatz et al. Science 345, 1153 (2014)



Bernd Gludovatz et al. Science 345, 1153 (2014)

Preparation of HEA



Arc melting



Induction melting

- 1. Excessive $Mn \rightarrow$ to compensate the loss of Mn by evaporation;
- 2. Pure $Zr \rightarrow$ to remove oxygen;
- 3. Iterative melting for 5 times \rightarrow to improve the homogeneity

Much time and energy consumption \rightarrow low efficiency

Challenge for preparation of HEA

- 1. Very different melting points (T_m) of elements \rightarrow How to depress the evaporation of low- T_m elements while assuring full melting of high- T_m elements?
- 2. How to reduce the oxidation of active elements.
- 3. How to avoid element segregation in a multi-element system?

I 元素	熔点/°C	i.
<u>V:</u>	1902	1
Nb:	2468	÷
Mo :	2610	1
Ta :	3017	1
W :	3422	- i

 $W_{20}Nb_{20}Ta_{20}Mo_{20}V_{20}$

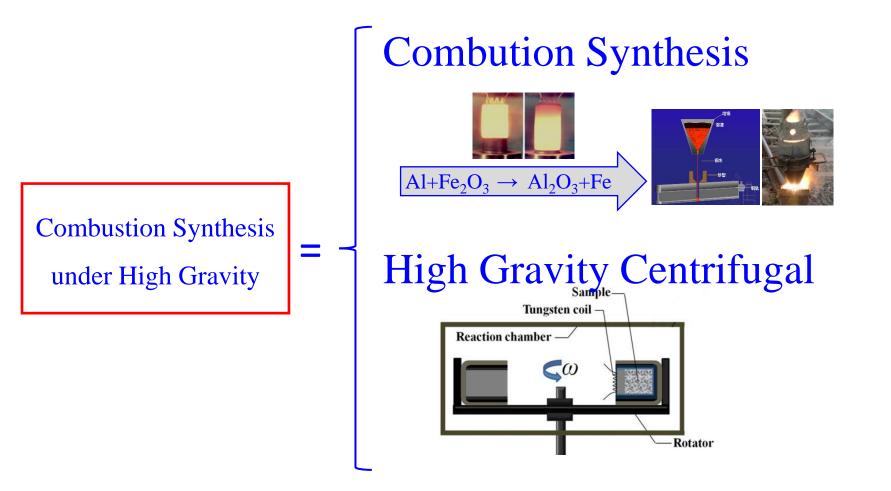
 $\Delta T_m = 1520$ °C

CrMnFeCoNi

│ 元素	熔点/℃ i
<u>Cr:</u>	<u>1857</u>
Mn:	1244
Fe:	1538 i
Co:	1495
Ni:	1453

 $\Delta T_m = 600°C$

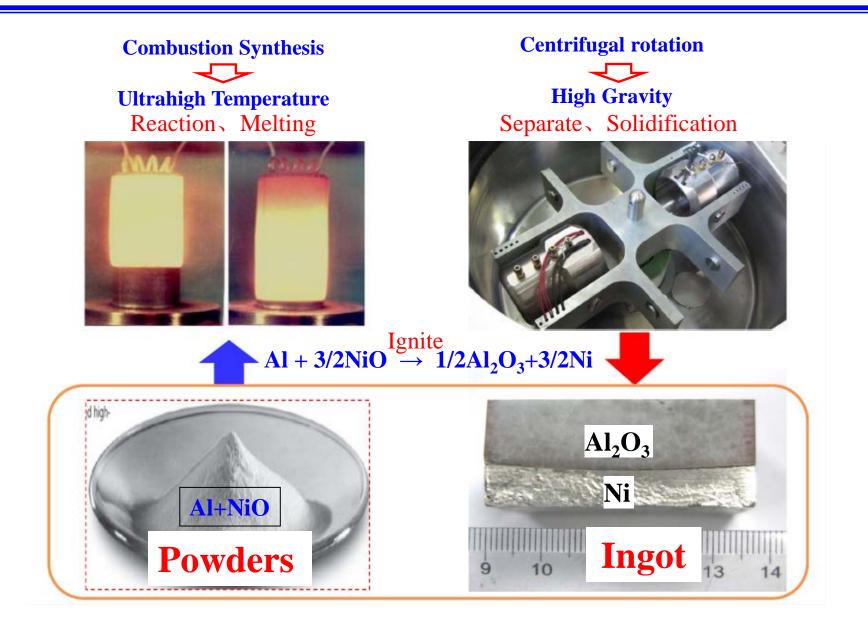
A new technique to prepare HEA: Combustion Synthesis under High Gravity Progress



Many Reaction Systems Available

Reaction system	Product	Melting point A	diabatic ter	nperature
$Al+3/2NiO \rightarrow 3/2Ni+1/2Al_2O_3$	Ni	1726	3524	
Al+8/3Co ₃ O ₄ \rightarrow 9/8Co+1/2Al ₂ O ₃	Co	1768	4181	
$Al+3/2CuO \rightarrow Cu+1/2Al_2O_3$	Cu	1357	3000	
$Al+1/2Cr_2O_3 \rightarrow Cr+1/2Al_2O_3$	Cr	2130	2831	
Al+1/2CrO ₃ \rightarrow 1/2Cr+1/2Al ₂ O ₃	Cr	2130	4000	> 20001/
$Al+3/10V_2O_5 \rightarrow 6/10V+1/2Al_2O_3$	v	2175	3785	>2800K
$Al+1/2WO_3 \rightarrow 1/2W+1/2Al_2O_3$	W	3680	4280	
$Al+1/2Fe_2O_3 \rightarrow Fe+1/2Al_2O_3$	Fe	1809	3622	
$14\text{Al}{+}3\text{CrO}_3 + 6\text{SiO}_2 \rightarrow 3\text{CrSi}_2 + 7\text{Al}_2\text{O}_3$	CrSi ₂	1748	3600	
$14\text{Al}{+}3\text{MoO}_3 + 6\text{SiO}_2 \rightarrow 3\text{MoSi}_2 {+}7\text{Al}_2\text{O}_3$	MoSi ₂	2293	3200	

Combustion Synthesis under High Gravity



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Facilities



Preparation of CrMnFeCoNi HEA

$Al+Cr_2O_3+Fe_2O_3+Co_2O_3+NiO+Mn \rightarrow Al_2O_3+\underline{CrMnFeCoNi}$



5 6 7 8 9 10

Outline

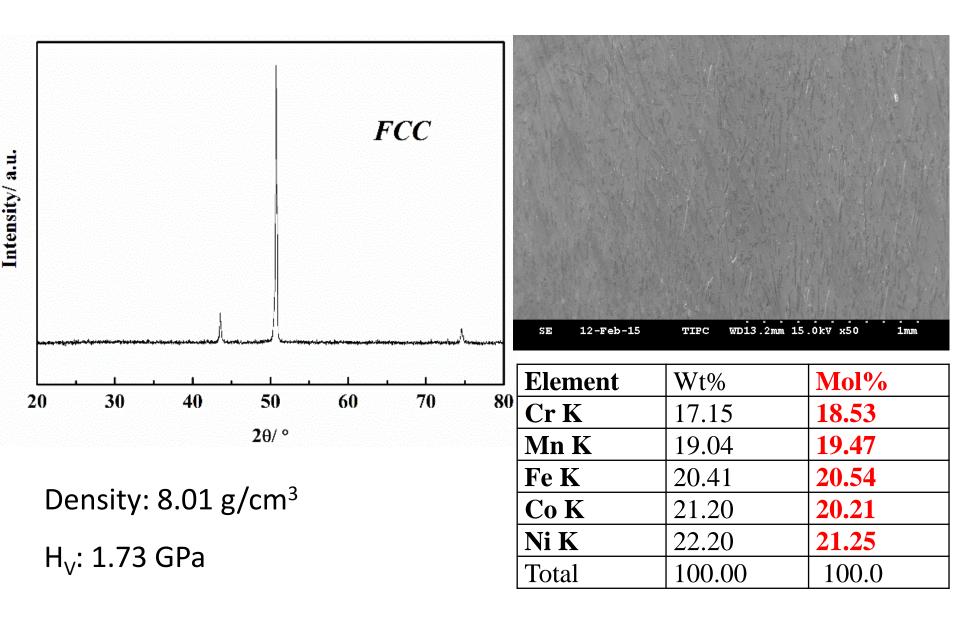
1. Introduction

2. Experimental

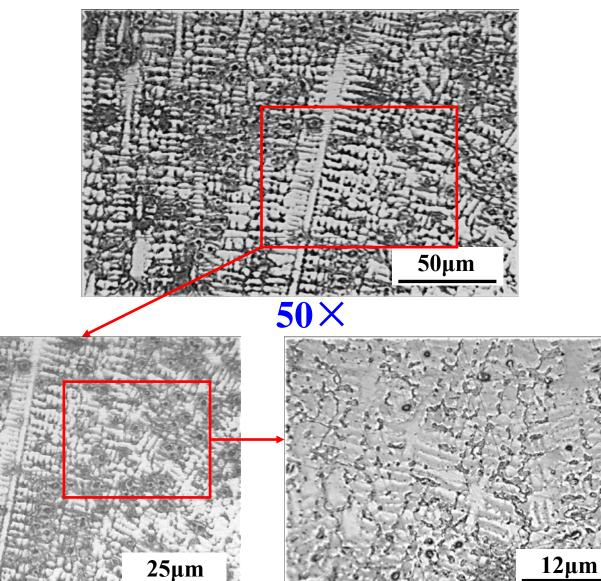
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XRD and SEM/EDS



Optical Micrographs



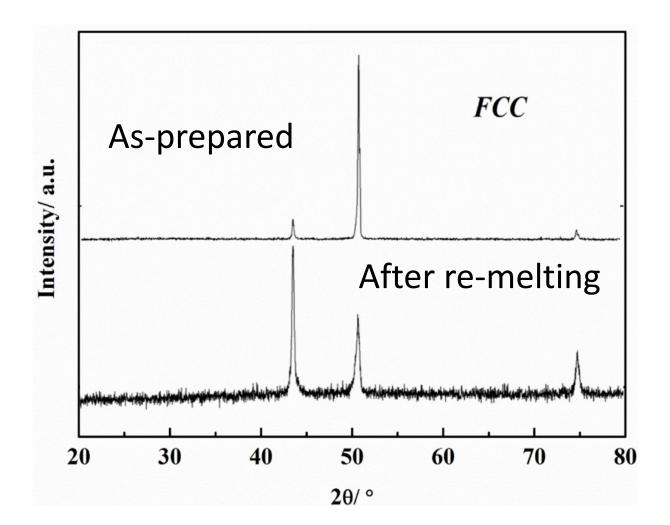
100×

5

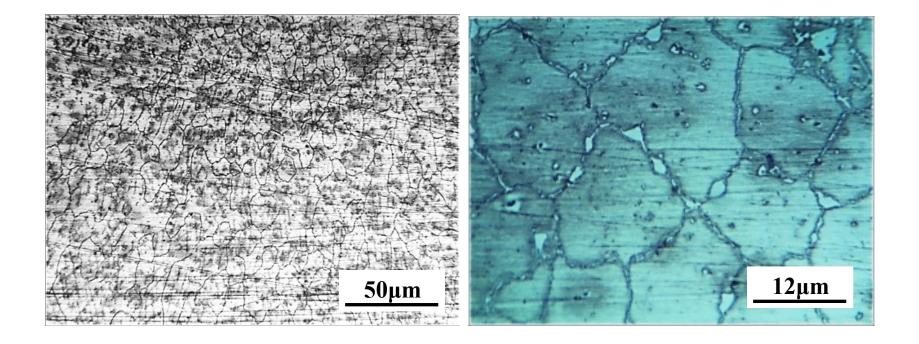
 $200 \times$

Strift Artific

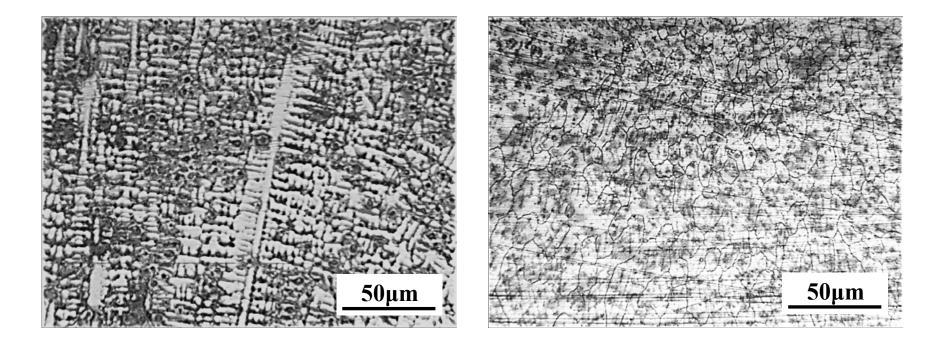
XRD after re-melting



Optical Micrographs after re-melting



Grain morphology changed after re-melting.

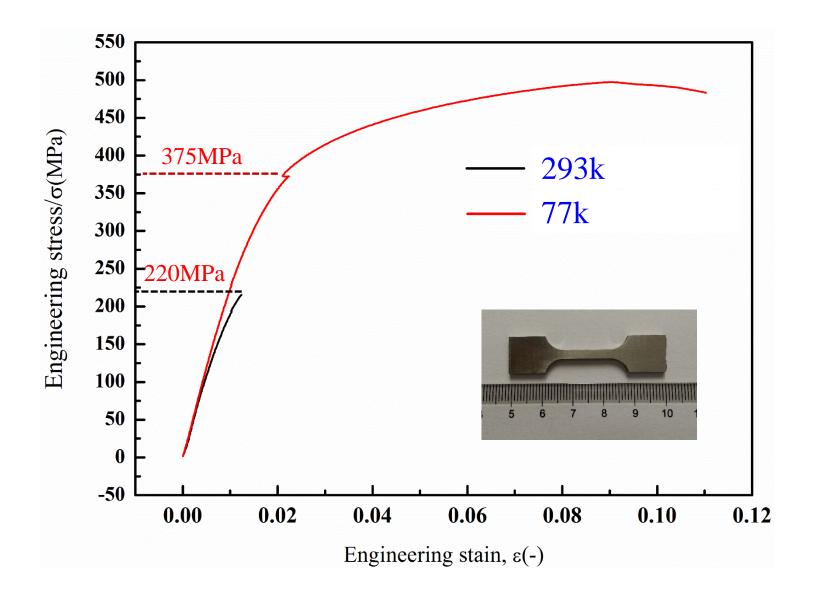


As-prepared

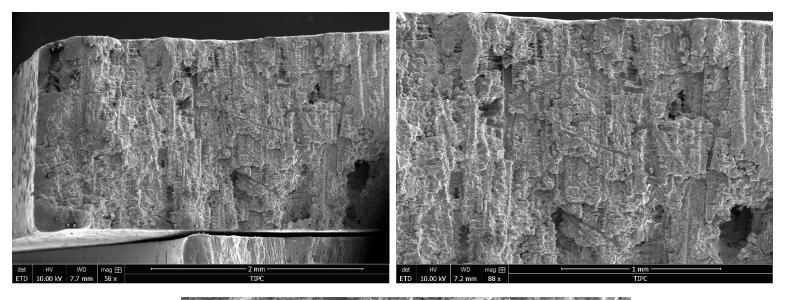
After re-melting

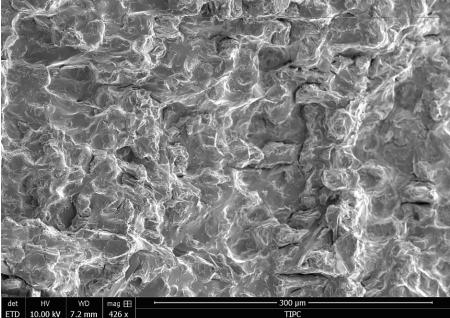
Dentritic \rightarrow Equiaxed

Tensile test of CrMnFeCoNi HEA

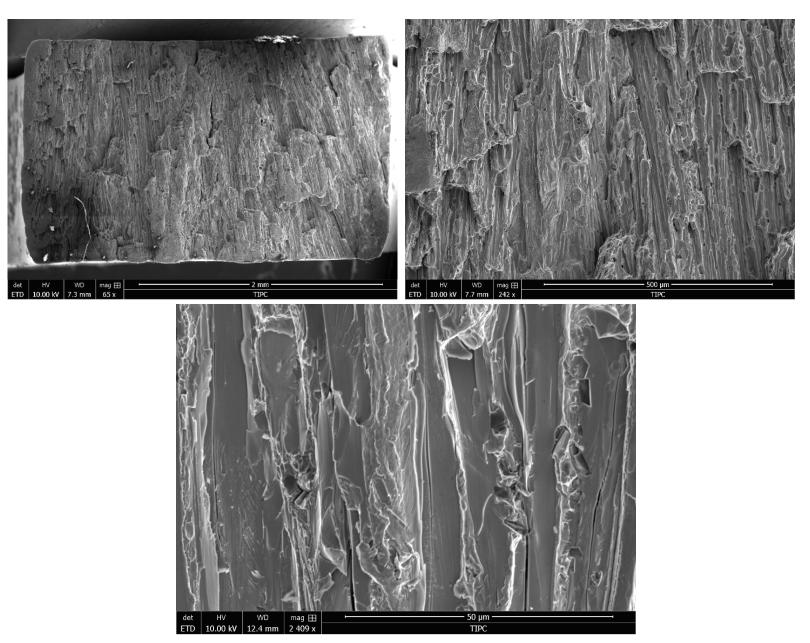


Morphology of fracture surface (fractured at 293K)

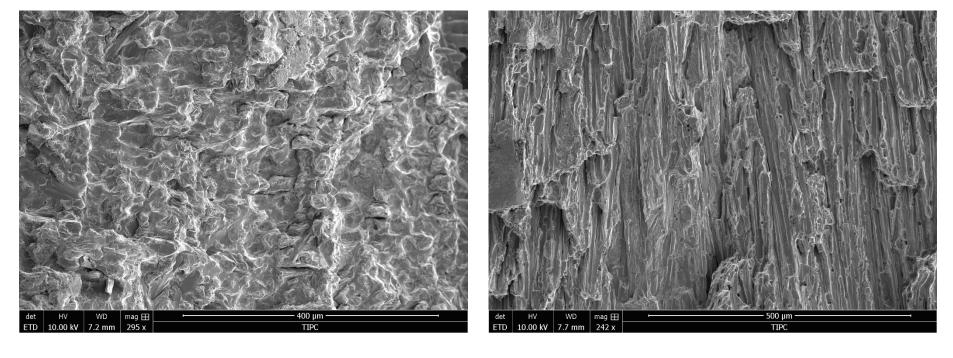




Morphology of fracture surface (fractured at 77K)



Comparison of morphology of fracture surface fractured at 293 K and 77 K





77 K

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Conclusions

- CrMnFeCoNi HEA was successfully prepared by combustion synthesis under high gravity, and showed single FCC lattice structure and dendritic grain morphology.
- 2. After re-melting, the CrMnFeCoNi HEA still kept the FCC structure, but the grain morphology became equiaxed.
- 3. The CrMnFeCoNi HEA exhibited brittle fracture at room temperature, but showed ductile behavior at 77 K, with much-improved strength and strain.

From the experiment results, combustion synthesis under high gravity may offer an alternative and more efficient way for preparing HEAs with promising cryogenic properties.

thermoelectric materials

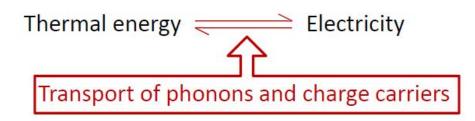
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Thermoelectric materials

Thermoelectric materials



Applications

Advantages

- 1. Do not consume fossil fuel
- 2. Do not produce CO₂
- 3. No moving parts, no noise
- 4. Pollution-free

A car with themoelectric waste heat recovery system BMW



A portable refrigerator with a thermoelectric cooling system

Mars Exploration Rover "**Curiosity**" with thermoelectric generator powered by Isotope heat source



The Eco Thermal Drive watch Citizen



Preparation of thermoelectric materials

Thermoelectric (TE) materials $ZT = \frac{\alpha}{T}$

$$T = \frac{\alpha^2 \sigma}{\kappa} T$$

α: Seebeck coefficient σ: Electrical conductivity κ: Thermal conductivity

Conventional methods for preparing bulk TE materials:

- **1.Growth from melt**
- 2.Sintering from powder

Long-time heat treatment by furnace

→ Much time and energy consumption

Q: How to reduce the porosity?

A: Depress the formation and accelerate the removal of gas bubbles.

Two approaches of modified CS:

1.Gas-pressure combustion synthesis

Clausius-Clapeyron Equation

$$\ln \frac{P_2}{P_1} = \frac{\Delta H_{_{V}}}{R} \left(\frac{1}{T_1} - \frac{1}{T_2}\right)$$

2.High-gravity combustion synthesis

$$V = 2/9 \cdot \rho g r^2 / \eta$$

Stokes' Equation

$$\begin{array}{ccc} \uparrow^{\mathbf{V}} & & \stackrel{\uparrow}{\underset{\mathbf{G'}}{\overset{\mathsf{F}}{\overset{\mathsf{F}}}}} \\ & & \stackrel{\bullet}{\underset{\mathbf{G'}}{\overset{\mathsf{F}}{\overset{\mathsf{F}}}}} \\ & & & \stackrel{\bullet}{\underset{\mathbf{G'}}{\overset{\mathsf{F}}{\overset{\mathsf{F}}}}} \\ \end{array}$$

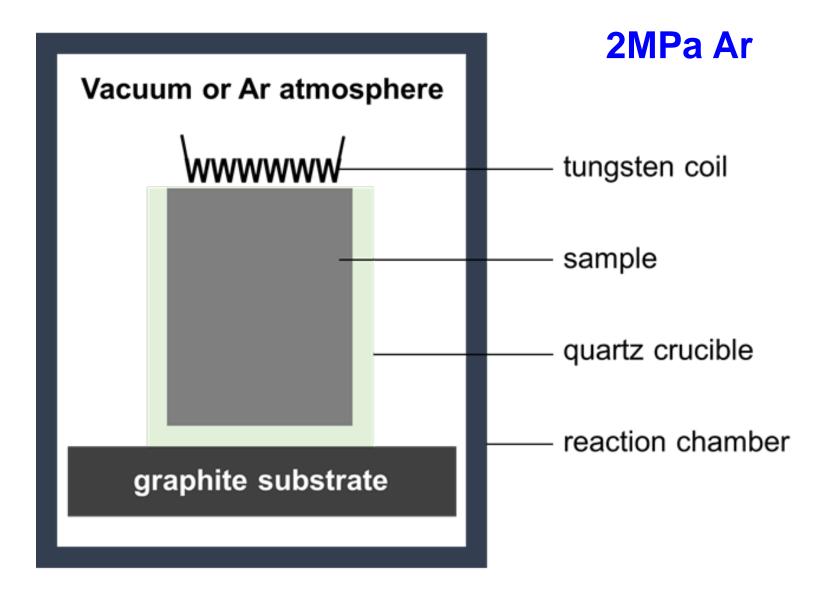


• Introduction

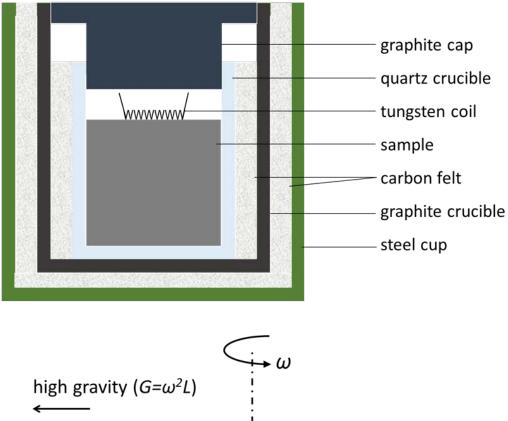
• Experimental

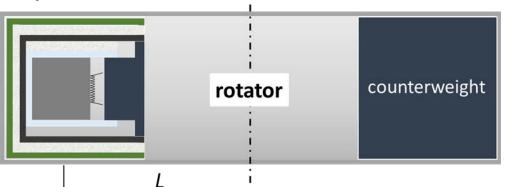
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Gas-pressure combustion synthesis



High-gravity combustion synthesis





G=800 g g=9.8 m/s²

Outline

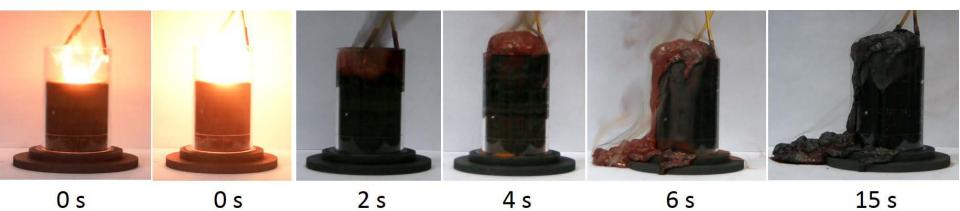
• Introduction

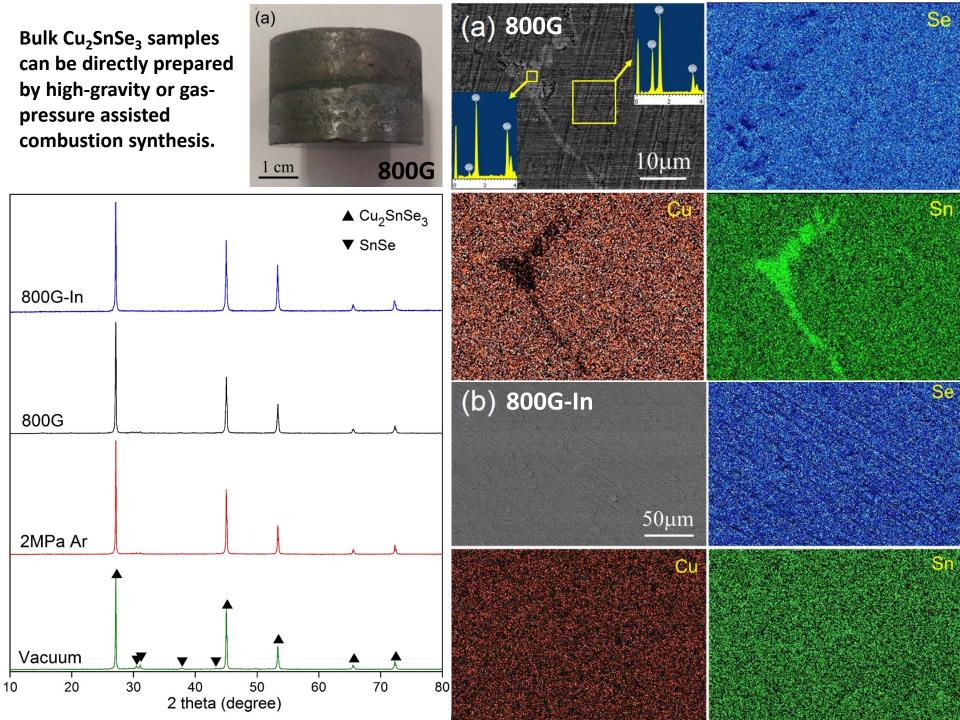
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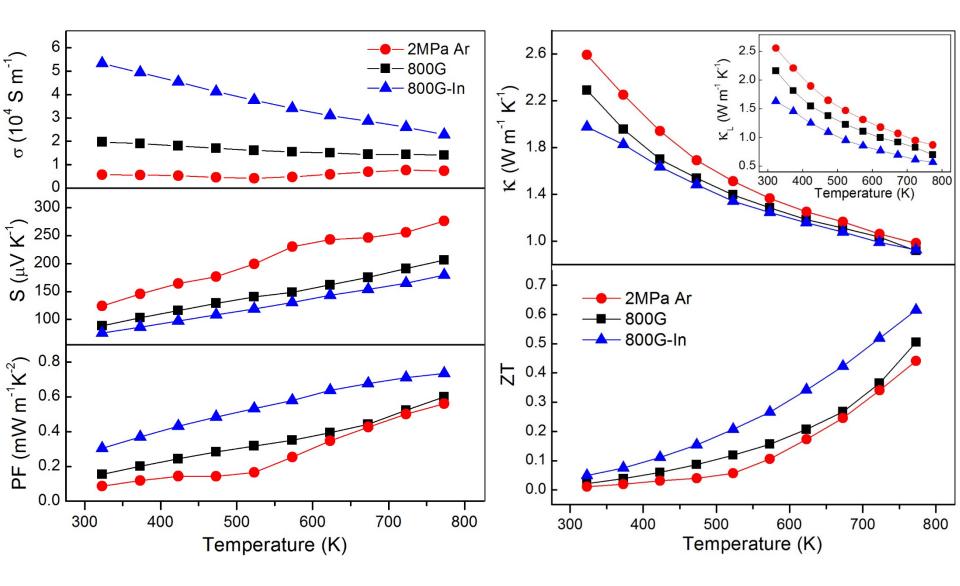
Reaction process in air

$2Cu + Sn + 3Se = Cu_2SnSe_3$



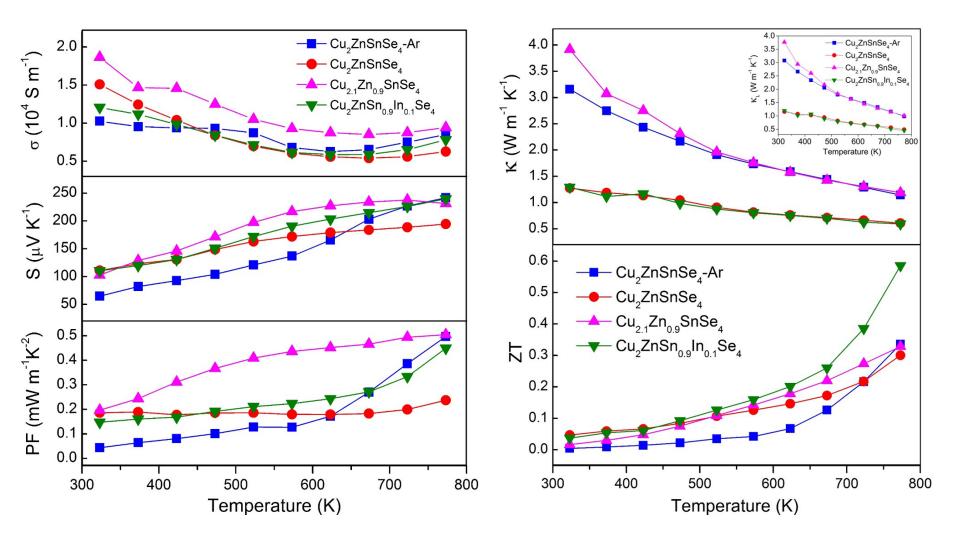


TE properties: comparable with sintered samples



2. $Cu_2ZnSnSe_4$ **2**Cu + Zn + Sn + **3**Se = $Cu_2ZnSnSe_4$

TE properties: comparable with sintered samples



Outline

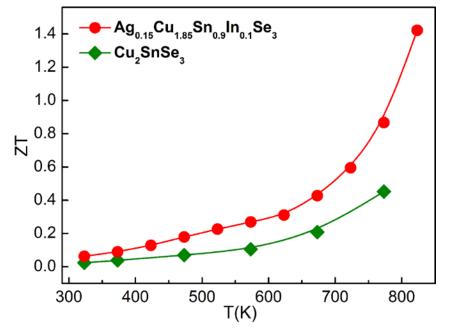
- Introduction
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Combustion synthesis of thermoelectric materials one-step, fast, furnace-free, scalable

• Combustion synthesis offers an alternative approach to the fabrication of thermoelectric materials with much reduced processing time and energy consumption.

• The synthesized Cu₂SnSe₃ and Cu₂ZnSnSe₄ samples show thermoelectric properties similar to those prepared by the conventional methods.

 The thermoelectric properties can be optimized by doping.



Thank you for your attention!